

Article

Highly Efficient Visible-Light-Induced Photocatalytic Activity of Fe-doped TiO₂ Nanoparticles

Paruchai Pongwan¹, Burapat Inceesungvorn², Khatcharin Wetchakun³,
Sukon Phanichphant⁴, and Natda Wetchakun^{1,*}

¹ Department of Physics and Materials Science, Faculty of Science, Chiang Mai University, Chiang Mai 50200, Thailand

² Department of Chemistry, Faculty of Science, Chiang Mai University, Chiang Mai 50200, Thailand

³ Nanoscience and Nanotechnology Program, Graduate School, Chiang Mai University, Chiang Mai 50200, Thailand

⁴ Materials Science Research Center, Faculty of Science, Chiang Mai University, Chiang Mai 50200, Thailand

E-mail: natda_we@yahoo.com*

Abstract. Bare TiO₂ and nominal 5.0 at% Fe-doped TiO₂ nanoparticles were synthesized by the modified sol-gel method. The samples were physically characterized in order to obtain the correlation between structure and photocatalytic properties by X-ray diffraction (XRD), scanning electron microscopy (SEM), transmission electron microscopy (TEM), Brunauer, Emmett and Teller (BET), and UV-vis diffuse reflectance spectrophotometry (UV-vis DRS). XRD results indicated that phase structures of bare TiO₂ and Fe-doped TiO₂ nanoparticles were the mixture of anatase and rutile phases. The content of rutile phase in 5.0 at% Fe-doped TiO₂ nanoparticles decreased. TEM images revealed that the shape of bare and 5.0 at% Fe-doped TiO₂ was almost spherical and the average particle size was in the range of 10–30 nm. Specific surface areas of the samples were found as 75 and 134 m²/g for bare TiO₂ and nominal 5.0 at% Fe-doped TiO₂, respectively. The results from UV-vis reflectance spectra clearly indicated the shift of absorption band edge towards visible region upon doping TiO₂ with iron. Photocatalytic activity of bare TiO₂ and 5.0 at% Fe-doped TiO₂ nanoparticles was examined by studying the mineralization of oxalic acid under visible light irradiation and the results clearly showed that Fe-doped TiO₂ sample exhibited higher activity than bare TiO₂.

Keywords: Nanoparticles, Fe-doped TiO₂, photocatalytic activity, modified sol-gel, visible light.

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1. Introduction

Recently, the rapid growth of industries leads to the expansion of environmental problems. One of the main environmental problems is water pollution from industries, agricultures, houses, and so on, which have contaminants such as organic and inorganic compounds. Recently, the new approach for wastewater treatment is an application of photocatalyst through advanced oxidation process. Titanium dioxide (TiO_2) is one of the most efficient photocatalyst due to its high stability, chemical inertness, non-toxicity, and low cost [1–2]. Generally, TiO_2 has band gap energy of 3.2 eV for the anatase phase and 3.0 eV for the rutile phase of which the absorption thresholds correspond to 380 and 410 nm, respectively [3]. The photocatalytic process requires the energy of light equal or greater than the band gap energy of TiO_2 for production of electron and hole. The generated electron and hole can then react with water molecules and hydroxyl ions, which subsequently produce hydroxyl radical known as a strong oxidant for organic pollutant degradation in water. Apart from the oxidation process, the hole and electron can recombine easily, thus resulting in very low photocatalytic efficiency [4]. Therefore, many attempts have been made to retard this electron-hole recombination such as modification of TiO_2 surface by doping with various cationic and anionic elements [5–7]. Surface modification using appropriate element dopant could reduce the recombination rate of electron-hole pairs and increase the photocatalytic efficiency of TiO_2 . Recently, many TiO_2 modification procedures have successfully shifted the photocatalytic activity of TiO_2 from the UV region to visible light region; therefore enhanced photocatalytic activity could be obtained as a consequence of more complete utilization of solar energy [8]. Narrowing of TiO_2 band gap can be acquired by substitutionally or interstitially doped with various transition metal ions [9]. Doping of the transition metal ions such as Fe, Cu, and V, and so on was reported to induce a red shift of TiO_2 absorption spectrum and the enhancement of photocatalytic activity was also observed. Instead of using an expensive noble metal dopant, visible light photoactivity of TiO_2 has also been found in the TiO_2 doped with cheaper transition metal ions such as Fe and Cu. Although, doping of transition metal ion could increase photocatalytic efficiency of TiO_2 and the activity of this material also depends on various parameters related to the preparation method and the physical properties of the obtained catalyst. Several synthesis methods of TiO_2 have been proposed in literatures including sol-gel [9], flame spray pyrolysis [10], precipitation [11], and hydrothermal method [12]. A well-known method for preparing metal ion doping of TiO_2 is based on the sol-gel method via hydrolysis mechanism. This method offers several advantages, including high homogeneity and high chemical purity. Hence, the sol-gel method is widely used to prepare nano-sized TiO_2 photocatalysts [13–15]. Furthermore, the sol-gel method has successfully been employed for preparation of transition metal-doped TiO_2 photocatalyst and the enhanced photocatalytic activity under visible light irradiation was previously been observed [16–18].

In this research, the modified sol-gel method was employed for synthesis of Fe-doped TiO_2 nanoparticles. Photocatalytic degradation activities of the synthesized particles were studied using oxalic acid as the model organic compound under visible light irradiation.

2. Experimental

2.1. Sample Preparation

Bare and Fe-doped TiO_2 were synthesized by using titanium tetraisopropoxide (TTIP) as the Ti-precursor. Iron (III) nitrate ($\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$) was used as the metal precursor. 20 ml titanium tetraisopropoxide was dissolved in 250 ml 6M nitric acid solution and mixed until a homogeneous solution was obtained. The mixture of TTIP and nitric acid solution was loaded into a cellophane membrane and suspended for 1h in a clear solution containing 0.001 M of ethylenediaminetetraacetic acid (EDTA) and 7 ml of ammonia solution (25 %). The solution of iron (III) nitrate containing iron amounts of 5.0 at% was then added into the cellophane membrane pouch. The advantage of using the cellophane membrane is to control the diffusion rate of hydrolysis and condensation reactions in order to obtain small catalyst particles. After the completion of the dialysis process, the suspension was centrifuged at 5000 rpm for 10 min, washed with deionized water and then dried in an oven at 60°C for 24h. The obtained powder was finally calcined in a furnace at a temperature of 400°C for 3h.

2.2. Sample Characterization

Crystal structure and crystallite size of the samples were determined by X-ray diffraction spectroscopy (XRD, Philips X' Pert MPD) with $\text{CuK}\alpha$ radiation and calculated by using Scherrer equation. Rietveld refinement was carried out to determine the phase composition of the prepared samples. Morphology and particle size were investigated by transmission electron microscopy (TEM, JEOL JEM-2010) and scanning electron microscopy (SEM, JEOL JEM-6335F). Elemental composition of the obtained catalyst was analyzed by energy dispersive X-ray spectroscopy with SEM instrument (SEM-EDS). The Brunauer Emmett and Teller (BET) adsorption-desorption of nitrogen gas for specific surface area determination at the temperature of liquid nitrogen was performed on autosorb-1MP-Quantachrome. UV-vis absorption spectra and band gap energy were studied by UV-vis diffuse reflectance spectrophotometry (Lambda 650S).

2.3. Photocatalytic Activity Studies

The suspension of bare TiO_2 and Fe-doped TiO_2 nanoparticles (1 g/L) were prepared in deionized water and the pH was adjusted to 3.0 using 0.2 M of perchloric acid. An impurity carbon burn-off step was firstly carried out by illuminating the photocatalyst suspension with a UVA lamp (Sylvania blacklight blue, 18W) at ambient condition until no further CO_2 generation was obtained. The suspension was then air-equilibrated before an injection of oxalic acid (500 μg of carbon). Prior to the illumination, the catalyst suspension containing oxalic acid equivalent to 500 microgram carbon was circulated through the photoreactor under the dark for 30 min to establish adsorption/desorption equilibrium. Photocatalytic reaction was then initiated by illuminating the suspension with a fluorescent lamp (Sylvania fluorescent lamp, 18 W) filtered with a Rosco[®] E-colour UV filter. The amount of carbon dioxide generated by the mineralization of oxalic acid was determined via conductivity measurement. The photocatalytic performance of the synthesized samples was analyzed through the rate of oxalic acid photomineralization.

3. Results and discussion

3.1. UV-vis Diffuse Reflectance Spectra and Band Gap Energy

The absorption spectra were obtained by analyzing the reflectance measurement with Kubelka-Munk emission function, given by the equation below:

$$F(R_\infty) = [1 - (R_\infty)]^2 / 2R_\infty \quad (1)$$

where $F(R_\infty)$ is proportional to the absorption constant of the material, an indicative of the absorbance of the sample at particular wavelength. Band gap of the sample was obtained from a plot between:

$$E_g = 1241 / \lambda_{\text{onset}} \text{ and } [F(R_\infty)h\nu]^{1/2} \quad (2)$$

where E_g is the band gap energy in eV, h is Planck's constant, and ν is the frequency.

Figure 1 shows the UV-vis spectra of commercial TiO_2 (Degussa P25), bare TiO_2 and 5.0 at% Fe-doped TiO_2 . The results clearly showed a shift of absorption band edge towards visible region upon doping TiO_2 with iron. This suggested that the excitation energy of nano-sized Fe-doped TiO_2 decreases upon iron doping as seen from the inset of Fig. 1. The absorption band edges of Degussa P25, bare TiO_2 and nominal 5.0 at% Fe-doped TiO_2 were found in the range of 380–450 nm, corresponding to the band gap energy of 3.25, 2.99, and 2.64 eV, respectively. According to the UV-vis results, it can be assumed that Fe-doping onto TiO_2 may enhance the visible-light absorption and improve the photocatalytic activity of TiO_2 .

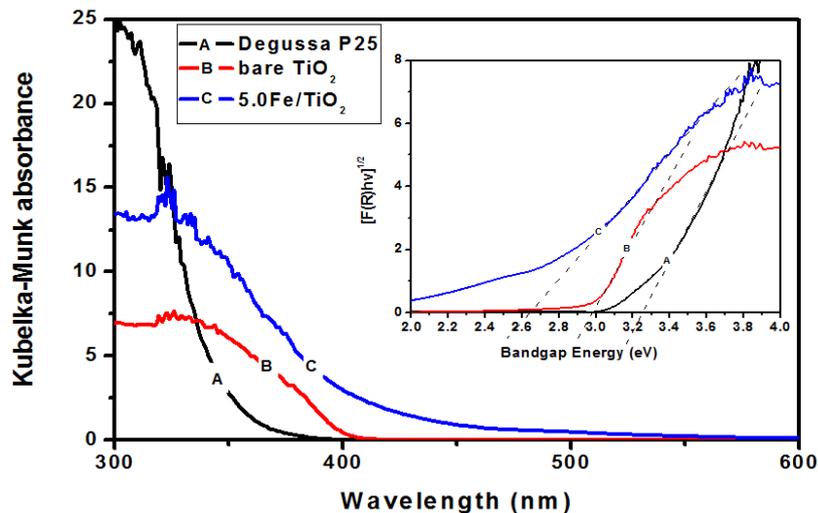


Fig. 1. Kubelka-Munk absorbance spectra of Degussa P25, bare TiO_2 and nominal 5.0 at% Fe-doped TiO_2 . The inset shows band gap energies of all samples.

3.2. XRD and BET Analysis

XRD patterns of bare TiO_2 and nominal 5.0 at% Fe-doped TiO_2 nanoparticles are shown in Fig. 2. Both samples presented in the mixed crystalline phases between anatase and rutile. The existence of both anatase and rutile phases in bare TiO_2 and Fe-doped TiO_2 nanoparticles can be identified by JCPDS file no. 21–1272 and no. 21–1276, respectively. An increase of anatase phase content was observed in Fe-doped TiO_2 sample. It was previously reported that excessive Fe doping disturbs the arrangement of TiO_2 octahedral units in rutile structure and hence distorts the overall crystallite structure [19]. The XRD peak of Fe dopant was not detected possibly because iron was doped in the range of very low concentration [20].

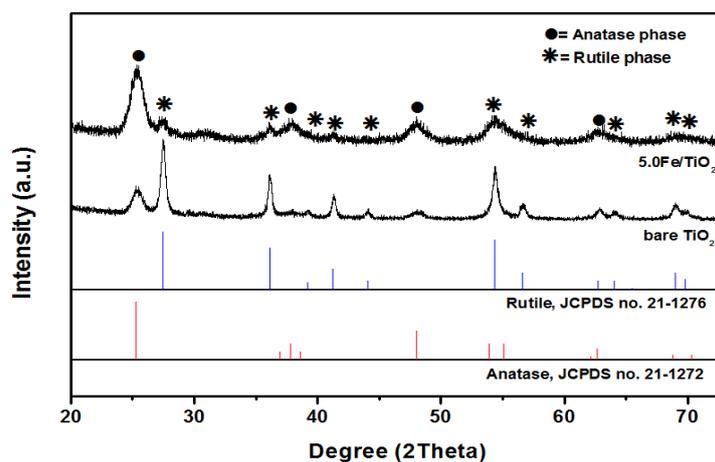


Fig. 2. X-ray diffraction patterns of bare TiO_2 and nominal 5.0 at% Fe-doped TiO_2 nanoparticles.

Phase composition, crystallite size, and BET specific surface area of bare TiO_2 and nominal 5.0 at% Fe-doped TiO_2 are presented in Table 1. It was found that the percentage of rutile phase in Fe-doped TiO_2 decreases upon iron doping. An average crystallite size of the samples has been calculated by Scherrer equation as illustrated below:

$$D = \frac{0.9\lambda}{\beta_{(\text{anatase,rutile})} \cos \theta} \quad (3)$$

where D is an average crystallite size, λ is the X-ray wavelength, θ is Bragg angle, and β is the half-width of the full maximum (HWM) of the main peak of anatase or rutile phases. The average crystallite size of

bare TiO₂ and Fe-doped TiO₂ samples was in the range of 8–9 nm for anatase phase and 17–20 nm for rutile phase. No iron-containing phase could be resolved in these patterns, which suggested insignificant iron segregation in Fe-doped TiO₂ nanoparticles. Peak position of (101) plane of anatase phase was shifted upon doping TiO₂ with iron, indicating the distortion of TiO₂ crystal lattice by iron dopant. TiO₂ lattice distortion was previously ascribed to the substitution of titanium ion with iron ion because the ionic radius of Fe³⁺ (0.64 Å) was slightly smaller than that of Ti⁴⁺ (0.68 Å) [21]. The substitution of Fe³⁺ for Ti⁴⁺ site in TiO₂ lattice is evidenced by the change in unit cell volume as reported in Table 2. According to Table 2, lattice parameters calculated from the (101) and (200) for anatase and (110) and (101) for rutile match very well with those of TiO₂ in JCPDS file no. 21–1272 and 21–1276 for anatase and rutile, respectively. A change in the unit cell volume of 5 at% Fe-doped TiO₂ compared with that of bare TiO₂ indicates that Fe³⁺ replaces Ti⁴⁺ in the TiO₂ lattice, and forms solid solution [22, 23]. BET specific surface area of bare TiO₂ and nominal 5.0 at% Fe-doped TiO₂ was 75 and 134 m²/g, respectively. The BET results showed that doping of TiO₂ with Fe results in an increase of specific surface area. Therefore, it is possible that the changes in particle size and specific surface area of Fe-doped TiO₂ nanoparticles are due to Fe³⁺ substituted for Ti⁴⁺ in TiO₂ lattice.

Table 1. Crystallite size, phase composition, and specific surface area of bare TiO₂ and nominal 5.0 at% Fe-doped TiO₂ nanoparticles.

Samples	Crystallite size (nm)		Phase composition (wt%)		Specific surface area (m ² /g)
	Anatase in (101) plane	Rutile in (110) plane	Anatase	Rutile	
Bare TiO ₂	9.0	17.9	33.40	66.60	75
5.0 at.% Fe-doped TiO ₂	8.1	20.1	89.99	10.01	134

Table 2. Unit cell parameters and unit cell volumes of bare TiO₂ and nominal 5.0 at% Fe-doped TiO₂ nanoparticles in comparison with the JCPDS file standard data of TiO₂.

Samples	Lattice parameters (Å)				Unit cell volume (Å ³)	
	Anatase		Rutile		Anatase	Rutile
	<i>a</i>	<i>c</i>	<i>a</i>	<i>c</i>		
TiO ₂ (JCPDS file no. 21-1272 and 21-1276)	3.7852	9.5139	4.5933	2.9592	136.31	62.43
Bare TiO ₂	3.7850	9.5135	4.5926	2.9589	136.29	62.40
5 at% Fe-doped TiO ₂	3.7880	9.3864	4.6404	2.9443	134.69	63.40

3.3. Morphology Characterization

The SEM images showed the particle size and morphology in a rough scale measurement. The morphology of bare TiO₂ and was found in an almost spherical particles with diameters ranging from 10–30 nm as shown in Fig. 3(a) and (b). The energy dispersive X-ray spectroscopy (EDS) analysis of 5.0 at% Fe-doped TiO₂ as shown in Fig. 3(c) indicated the presence of Ti, O and Fe in Fe-doped TiO₂ nanoparticles.

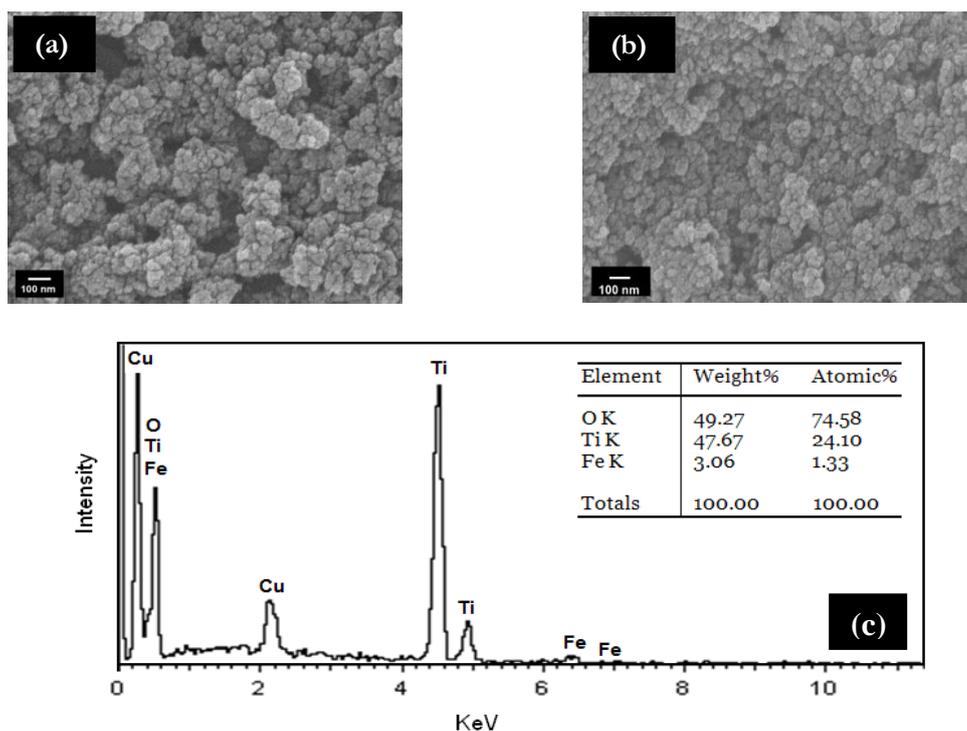


Fig. 3. SEM images of (a) bare TiO₂, (b) 5.0 at% Fe-doped TiO₂ and (c) EDS spectra of 5.0 at% Fe-doped TiO₂.

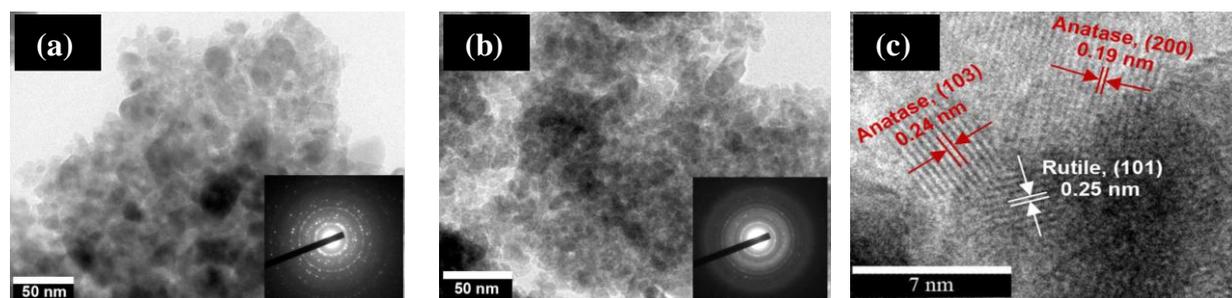


Fig. 4. TEM and SAED images of (a) pure TiO₂, (b) 5.0 at% Fe-doped TiO₂, and (c) lattice fringe image of 5.0 at% Fe-doped TiO₂.

Although SEM images gave a rough estimation of particle sizes, TEM images can reveal internal structure and a more accurate measurement of particle size and morphology. The TEM images and selected area electron diffraction (SAED) pattern of bare TiO₂ and 5.0 at% Fe-doped TiO₂ are shown in Fig. 4(a) and (b), respectively. The TEM images of both bare TiO₂ and Fe-doped TiO₂ revealed an almost spherical shape with average particle size in the range of 10–30 nm. SAED pattern confirmed the mixed-phase structure according to the set of concentric rings which have been indexed as various planes of anatase and rutile structures. Fig. 4(c) shows lattice fringes of 5.0 at% Fe-doped TiO₂ sample. The distance between each lattice fringes were measured and identified as the crystal structure of TiO₂ because the *d*-spacing of TiO₂ lattice plane from JCPDS file matches very well with the width of lattice fringe from TEM images. Consequently, the fringe widths of 0.24 nm and 0.19 nm can be used to confirm the dominance of (103) and (200) planes of anatase structure. On the contrary, the fringe width of 0.25 nm confirmed the dominance of (101) plane of rutile structure.

3.4. Photocatalytic Activity of Oxalic Acid Mineralization

Photoactivity of the prepared samples was evaluated by the photocatalytic oxidation reaction of oxalic acid under visible light irradiation as shown in Fig. 5. As a comparison, the photolysis of oxalic acid was also carried out at the same condition but without any catalyst. It was found that only *ca.* 5.62% oxalic acid is degraded under the same irradiation time. It is clearly seen that Fe-doped TiO₂ displayed much higher photoactivity than bare TiO₂ under visible light illumination. The photocatalytic mineralization efficiency was increased to 77 % with the use of 5.0 at% Fe-doped TiO₂. This is possibly due to the visible-light absorption ability of Fe-doped TiO₂ as evidenced by UV-vis diffuse reflectance spectra. An increased photocatalytic activity obtained in the case of Fe-doped sample has also been ascribed to Fe particles acting as an electron trap, retarding the electron-hole recombination process, and thereby, promoting the photocatalytic activity [24].

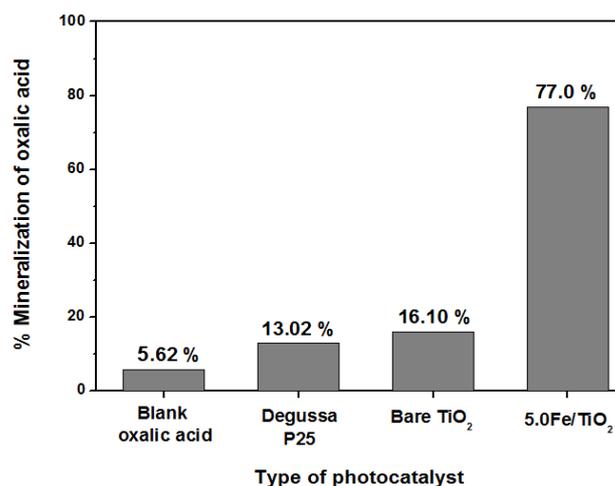


Fig. 5. Mineralization of oxalic acid under visible light illumination with an irradiation time of 180 min over P25, bare TiO₂ and 5.0 at% Fe-doped TiO₂ compared with oxalic acid photolysis.

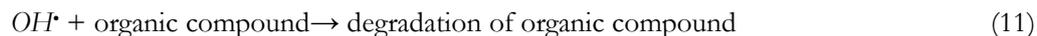
It has been clearly established that the photocatalyzed mineralization of organic compound in solution is initiated by photo-excitation of the metal oxide semiconductor, followed by the generation of an electron-hole pair on the surface of catalyst (Eq. (4)). Direct oxidation of oxalic acid (Eq. (5)) can also occur due to high oxidative potential of the generated hole (h^{+}_{VB}) in the catalyst. Very reactive hydroxyl radicals can also be formed either by the decomposition of water (Eq. (6)) or by the reaction of hole with OH⁻ (Eq. (7)). The hydroxyl radical known as an extremely strong, non-selective oxidant can lead to the partial or complete mineralization of several organic chemicals.



Electron in the conduction band (e^{-}_{CB}) on the catalyst surface can reduce molecular oxygen to superoxide anion (Eq. (8)). This radical, in the presence of organic scavenger, may form organic peroxide (Eq. (9)) or hydrogen peroxide (Eq. (10)).



Electrons in the conduction band are also responsible for the production of hydroxyl radicals, species which have been indicated as the primary cause of organic compound degradation (Eq. (11)).



4. Conclusions

Nominal 5.0 at% Fe-doped TiO₂ nanoparticles have been successfully synthesized by the modified sol-gel method. The XRD patterns showed the presence of mixed anatase and rutile phases of TiO₂ structure. The particle size and morphology investigated by TEM images revealed an almost spherical shape with particle size in the range of 10–30 nm. Photocatalytic activities of bare TiO₂ and Fe-doped TiO₂ samples were investigated by visible light-induced mineralization of oxalic acid in aqueous solution. The results clearly showed that doping of Fe could greatly enhance the photocatalytic activity of TiO₂ nanoparticles for oxalic acid mineralization. This enhanced photocatalytic activity is possibly due to the visible-light absorption ability of Fe-doped TiO₂ as observed from UV-vis spectra. In addition, an increase of specific surface area, a shift of absorption band edge toward visible region, and a decrease of TiO₂ band gap energy observed in our BET and UV-vis study could be the possible reasons explaining for an increased photocatalytic efficiency of the as-synthesized Fe-doped TiO₂ sample.

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